

Size-Dependent Bond Dissociation Enthalpies in Single-Walled Carbon Nanotubes

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We report the bond dissociation enthalpy (BDE) and the local electronic properties of Single-Walled Carbon Nanotubes (SWCNT) using density functional theory. Our analysis shows that there is a strong size-dependence of the BDE of these SWCNTs, which is inversely proportional to the radius-squared ($1/r^2$) and the length ($1/l$) of SWCNT. We derive quantitative relationships from which the BDE can be calculated as a function of size and radius of the SWCNT. We find that the BDE of SWCNT outside the size-dependent region is about 480 kJ mol⁻¹, which can be used for thermochemical calculations.

1. Introduction

Single-Walled Carbon Nanotubes (SWCNT) are used in many nanotechnology and commercial applications because of their unusual optical, electrical, mechanical and chemical properties.[1] Some of the wide range of applications include microelectronics, medicinal therapy, biosensors, gas-sensors, computing and one of their broadest use is in composite materials, to enhance their mechanical strength.[2] SWCNTs can produce materials such as nanosheets[3] and nanofibers[4,5] with very strong elastic properties. SWCNTs filled with transition metal chains have been suggested as one-dimensional nanocables.[6] These applications rely in part on the strong elastic properties, the thermodynamic stability and inertness of SWCNTs and therefore, understanding these properties is important on a

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microscopic level. There are recent theoretical studies on structural and electronic effects of finite size zigzag or armchair carbon nanotubes of various diameters and lengths[7], their charge polarisation[8], the effect of doping[9] and the role of tube chirality on the diffusion of water.[10] There have also been studies of the properties of hybrid SWCNT with other materials, mostly carbon materials, transition metals and even confined water molecules.[11] However, there are relatively few computational studies of the thermodynamic of CNTs as a function of their radius, diameter, chirality, functionalisation and doping. To the best of our knowledge there is currently no other study of the size-dependent BDE of CNTs.

Previously, computational studies have shown that the elastic constant of zig-zag and arm-chair CNTs is smaller than that of graphene.[12] The bandgap of zig-zag (3n,0) CNTs converges from 1.6 eV at a length of 3 Å to 0.8 eV at a length of 9 Å.[12] The elastic properties (Young's, shear and bulk modulus) of functionalised CNTs have been studied with periodic density functional theory (DFT) calculations for functionalised (-NH, -NH₂, -CH₂, -CH₃, -OH) organic fragments.[13] The decrease of the elastic properties is roughly proportional to the strength of the binding of the functional groups and it can be 30% for high concentrations of adsorbates.[13] Generalized gradient approximation (GGA) based DFT calculations show that the Young's modulus of SWCNT increase as a function of their radius from 0.946 TPa for the CNT(3,3) to 1.040 TPa of the CNT(9,9)[14], which indicates that the C-C bond becomes stronger as the diameter of the CNT increases from 0.2 nm to 0.6 nm.[14] Furthermore, hybrid exchange and correlation functional based calculations (i.e. B3LYP/6-31G) of Raman and IR spectra show that in 14 zig-zag SWCNTs there is an increasing frequency in their A_{2u} (bending along radial axis) and E_{1u} (twisting along radial axis) bands, as a function of the tube diameter[15], which suggests that the elastic properties scale linearly with tube diameter.[15] The adsorption of H₂ has been studied on the CNT(10,0) using GGA DFT calculations, which found that the barrier for dissociative chemisorption is 79 kcal mol⁻¹ and that the reaction does not happen spontaneously at 0 K.[16] Therefore, the use of dopants such as Li⁺ ion is necessary to use CNTs as hydrogen storage materials.[17] Energetic materials for propellants can be stored inside CNTs and are stabilised due to charge transfer from the CNT to the molecule.[18] Although these computational studies have addressed how the elastic and therefore the IR and Raman spectroscopic properties change as a function of the tube diameter and length, they have

not given an explanation of the observed trends on the microscopic level and on the basis of the BDE of SWCNTs.

In this periodic DFT study we report the BDE and electronic properties of SWCNTs with range of lengths and diameters. In particular, we have studied the size-dependent bond dissociation enthalpy of the C-C bond (BDE_{CC}) and the partial-density-of-states (PDOS) of various arm-chair and zig-zag SWCNTs in the length range of 3.6 nm and the radius range of 1.0 nm in order to explain the size-dependent effect of the BDE and consequently, the elastic properties of SWCNT.

2. Computational Methods

2.1 DFT calculations

We performed periodic DFT calculations using the VASP 5.4.1 code.[19,20] The projector augmented-wave method has been used to represent the core states. [21,22] Exchange and correlation (XC) effects were considered within the generalized gradient approximation (GGA) using the revised Perdew-Burke-Ernzerhof (revPBE) XC functional. [23] The SWCNT, were optimised inside an orthorhombic cell with a 20 Å vacuum gap between the CNTs and an axial vacuum gap of 15Å as shown in Fig. 1.

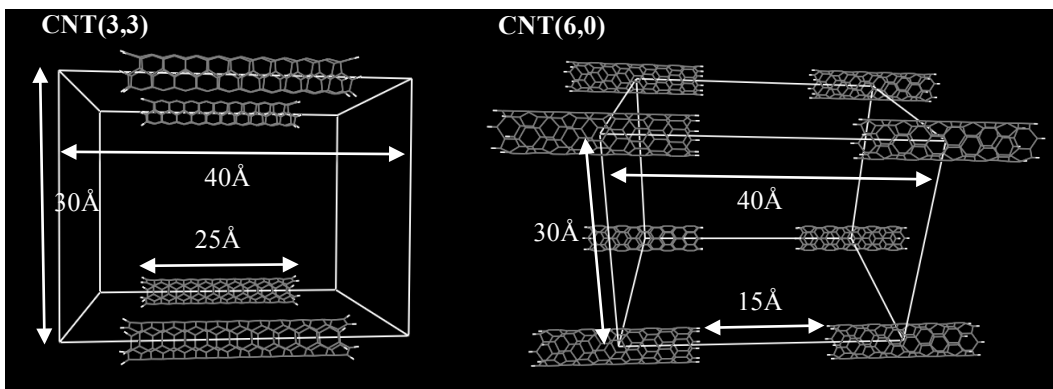


Fig. 1. The periodic unit cell of SWCNT(3,3) and SWCNT(6,0). The SWCNT length is 25 Å. For the 30 Å and 35 Å in length SWCNT, the unit cell width was 45 Å and 50 Å, respectively.

We have used a Γ -point[24] centered 2x2x2 Monkhorst-Pack grid for all calculations but the convergence of the energy was tested also with a 3x3x3 MP grid which gave energies to within 0.0001 eV. The coordinates of the SWCNT were generated in Nanotube Modeler[25] and

then rotated and translated to the highest space group using Materials Studio.[26] The corresponding point and space groups of each SWCNT are tabulated in Table 1. The dangling bonds in all SWCNT were saturated with one hydrogen atom per carbon atom. The cut-off energy for the energy of the planewaves was 600 eV. Geometry optimizations were performed with a residual force threshold on each atom of 0.01 eV Å⁻¹ using the conjugate-gradient algorithm. The convergence criterion for electronic relaxation was 10⁻⁴ eV. The initial charge density was obtained by superposition of atomic charges. Dispersion corrections were included via the zero-damping DFT-D3 correction method of Grimme as implemented in VASP.[27]

2.2 Calculation of bond dissociation enthalpies

The change in enthalpy for the bond formation of an SWCNT from its atoms was calculated based on the following equation,

$$\Delta H \cong E_{SWCNT} - n_C E_C - n_H E_H + n_C ZPV_C \quad (1)$$

, where n_C and n_H is the number of C and H atoms, respectively. E_{SWCNT} , E_C and E_H are the total energies of the SWCNT, of an isolated carbon atom in its triplet state (-1.626 eV) and a hydrogen atom in its doublet state (-1.178 eV) and the zero-point vibrational energy per carbon atom (ZPV_C) derived from a full electron B3LYP/STO-3G(d,f) calculation of the ZPV_C of circum-circum-coronene (C₉₆H₁₂) given by,

$$ZPV_C \cong \frac{ZPV(C_{96}H_{24})}{n_C} \quad (2).$$

The enthalpy of atomization of the SWCNT was calculated based on the following relationship,

$$\Delta H_{atom} = n_{CC} BDE_{CC} + n_{CH} BDE_{CH} \quad (3)$$

, where BDE_{CH} and BDE_{CC} are the average bond-dissociation energies of the C-H bond and C-C bond, respectively. The change in enthalpy for the bond formation of the SWCNT is equal to the negative atomization enthalpy,

$$\Delta H \cong -\Delta H_{atom} \quad (4).$$

Therefore, after the combination of equations 1, 3 and 4, the average BDE per C-C bond in an arbitrary SWCNT becomes,

$$BDE_{CC} \cong \frac{1}{n_{CC}} [-E_{SWCNT} + n_C E_C + n_H E_H - n_C ZPV_C - n_{CH} BDE_{CH}] \quad (5)$$

The BDE_{CH} was estimated based on the energy required to dissociate a hydrogen atom from coronene ($R=C_{24}H_{12}$) forming the corresponding radical ($R^{\bullet}=C_{24}H_{11}^{\bullet}$) given by,

$$BDE_{CH} = E_{RH} - E_{R^{\bullet}} - E_H + ZPV_{CH} \quad (6).$$

For the average BDE_{CH} of coronene we have used the value 454 kJ mol^{-1} , which was calculated at B3YP/cc-pVDZ(5d,7f) level of theory.

The ZPV of C-H was calculated with the following equation,

$$ZPV_{CH} = ZPV_{RH} - ZPV_{R^{\bullet}} \quad (7)$$

,where ZPV_{RH} and $ZPV_{R^{\bullet}}$ are the zero-point vibrational energies of coronene and coronene radical in its triplet state calculated at B3LYP/cc-pVDZ(5d,7f), respectively.[28]

3. Results and Discussion

3.1 Size-dependent properties of SWCNT as a function of their length

To the best of our knowledge, the trends of the bond dissociation enthalpy per C-C bond (BDE_{CC}) as a function of SWCNT length and radius, have not been previously reported. We have used equations 1-7 to calculate this thermodynamic property for various arm-chair and zig-zag SWCNT in which the dangling bonds were saturated with H-atoms. These results are tabulated in table 1 along with the chirality, point group (P.G.), symmetry group (S.G.), length (l), radius (r), number of C-C bonds (n_{CC}) and the number of C and H atoms (n_C , n_H).

Table 1. Chirality, point group (P.G.), symmetry group (S.G.), length (l), radius (r), number of C-C bonds (n_{CC}), the number of C (n_C) and H (n_H) atoms, length (l), radius (r) and the bond dissociation enthalpy per C-C bond is (BDE_{CC}) of the various arm-chair and zig-zag SWCNTs. Also shown the average BDE of C-C and C=C. [29]

Chirality	P. G.	S. G.	l (Å)	r (Å)	n _{CC}	n _C	n _H	BDE _{CC} (kJ/mol)
(6,0)	D _{6h}	PMMM	26.480	4.861	210	144	12	456.2
(7,0)	D _{7h}	PMM2	26.499	5.484	245	168	14	462.1
(8,0)	D _{8h}	P4/MMM	26.494	6.403	280	192	16	465.8
(9,0)	D _{9h}	PMM2	26.524	7.061	315	216	18	468.7
(10,0)	D _{10h}	PMMM	26.509	7.956	350	240	20	470.6
(3,3)	D _{3h}	PM	26.352	4.009	183	126	12	450.4
(4,4)	D _{4h}	P4/M	26.627	5.551	244	168	16	463.4
(5,5)	D _{5h}	PM	26.651	6.860	305	210	20	469.4
(6,6)	D _{6h}	P2/M	26.644	8.254	366	252	24	472.6
(7,7)	D _{7h}	PM	26.627	9.605	427	294	28	474.6
(3,3)	D _{3h}	PM	11.585	4.009	79	54	12	428.4
(3,3)	D _{3h}	PM	16.507	4.009	114	78	12	439.4
(3,3)	D _{3h}	PM	21.430	4.009	149	102	12	445.2
(3,3)	D _{3h}	PM	26.352	4.009	183	126	12	450.4
(3,3)	D _{3h}	PM	31.275	4.009	218	150	12	451.3
(3,3)	D _{3h}	PM	36.197	4.009	253	174	12	452.8
(6,0)	D _{6h}	P2/M	9.052	4.700	70	48	12	425.3
(6,0)	D _{6h}	P2/M	13.315	4.700	105	72	12	441.1
(6,0)	D _{6h}	P2/M	17.578	4.700	140	96	12	448.9
(6,0)	D _{6h}	P2/M	21.841	4.700	175	120	12	453.5
C-C	n/a	n/a	n/a	n/a	n/a	n/a	n/a	377
C=C	n/a	n/a	n/a	n/a	n/a	n/a	n/a	728

In order to study the BDE_{CC} as a function of SWCNT length we have chosen the zig-zag SWCNT(6,0) and arm-chair SWCNT(3,3), which have about the same radius, 4.861 Å and 4.009 Å, respectively, and had a relatively small number of carbon atoms so that longer tube lengths could be simulated. These SWCNT have relatively small diameter and therefore systems in which these have lengths as 4 nm could be studied. The bond dissociation enthalpy per C-C bond is proportional to 1/length of the CNT as this is shown in Fig. 2. This means that as the length of the SWCNT increases the BDE_{CC} becomes larger. For SWCNT that have longer lengths, there should be a convergence of the BDE_{CC} to a constant value, which however cannot be observed for CNT lengths < 4nm. In contrast we observe that the BDE_{CC} changes between 425 - 454 kJ mol⁻¹, corresponds to 6.3 %. We suggest that changes in the elastic properties of CNTs for

lengths $< 4\text{nm}$ could be as high as 6.3%, which means that size-dependent elastic properties have to be considered to accurately access such properties. This noticeable variation of the BDE_{CC} also suggests size-dependence of the thermodynamic properties for SWCNT that have lengths below 4 nm. This size-dependence of the properties should be observable in measurements of the elastic properties of SWCNT, such as Young's modulus, the tensile strength and the bending modulus, all of which are dependent on the bond strength of the C-C atoms within the carbon nanotube framework. Therefore, longer SWCNTs have in principle stronger mechanical properties such as Young's modulus and bending modulus and tensile strength as we have previously shown that there is linear correlation between the Young's modulus and the binding energy in graphene nanoribbons.[30] SWCNT are known to have the ability to withstand changes in length when under lengthwise tension or compression, which is a direct result of their high Young's modulus. The broad range of BDE_{CC} found in this study indicates that also there should be a broad range for the elastic properties of SWCNTs and MWCNTs. Analysis of the stress-strain curves of individual Multi-Walled Carbon Nanotubes (MWCNT) reveals Young's modulus of the outermost SWCNT in the range of 270 to 950 GPa measured with a “nanostressing stage” located within a scanning electron microscope.[31] The tensile strength of this layer varied from 11 to 63 GPa.[31] The wide range of Young's modulus and tensile strength in MWCNT is evident from the broad range of BDE_{CC} which were found to be in the range 428 kJ mol^{-1} - 453 kJ mol^{-1} and 425 kJ mol^{-1} - 454 kJ mol^{-1} for the CNT(3,3) and CNT(6,0), respectively. These CNTs had lengths that ranged between 1 to 4 nm and larger values for the BDE_{CC} are expected for longer SWCNT. By setting the length in the equation of the line in Fig. 1 to infinity we get a value of 463 kJ mol^{-1} and 480 kJ mol^{-1} for the CNT(3,3) and CNT(6,0), respectively. These BDE values are not identical as there is a 0.86 \AA radius difference between the two SWCNTs, and we show in the next section that there is a considerable radius dependence of the BDE of CNTs. Currently BDE are not available in the literature, however we can compare to BDE of organic molecules. In particular, we find that the BDE of SWCNT outside the size-dependent region is about 480 kJ mol^{-1} which is higher than that of the C-C bond, 377 kJ mol^{-1} and lower than that of the C=C bond, 728 kJ mol^{-1} , suggesting a partial double bond in SWCNTs.[29]

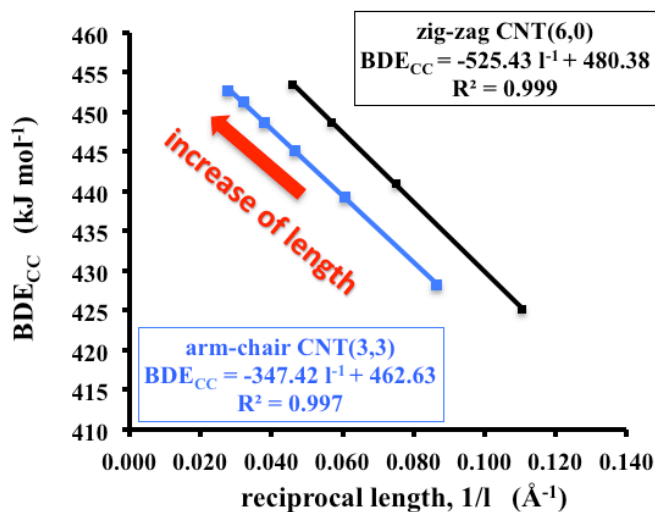


Fig. 2. Bond dissociation enthalpy per C-C bond as a function of the reciprocal CNT length. The length of the CNT(6,0) and CNT(3,3) ranged between 9.052 Å - 21.841 Å and 11.585 Å - 36.197 Å, respectively.

In Fig. 3 the converging trend of the BDE_{CC} can be seen. It is intriguing that even for 3 nm long CNTs the BDE_{CC} has yet to converge, which suggests that there is a size-dependence that governs this thermodynamic property. The fit to the data points is an equation of the form $BDE_{CC} = A - B/l$, where A and B are constants. This equation yielded residual-sum-of-squares (RSS) of 0.13 and 0.06, which indicates a good fit of the calculated data-points. By setting $l = \infty$ for the fit equation the BDE_{CC} of an infinite long CNT becomes 465 and 474 kJ mol⁻¹ for SWCNT(3,3) and SWCNT(6,0), respectively, values that are close to the results obtained earlier from the linear extrapolation. From Fig. 3 we estimate the length at which the BDE_{CC} converges, which is 8.3 nm and 8.7 nm for SWCNT(3,3) and SWCNT(6,0), respectively. This suggests that in SWCNT and MWCNT changes of the mechanical properties are expected for lengths < 9 nm and that only CNTs with dimensions > 9 nm should have isotropic thermodynamic and elastic properties (i.e constant as a function of size). This is quite important in the correct design of nano-, nano-sensors and nano-mechanical resonators that utilise CNTs.

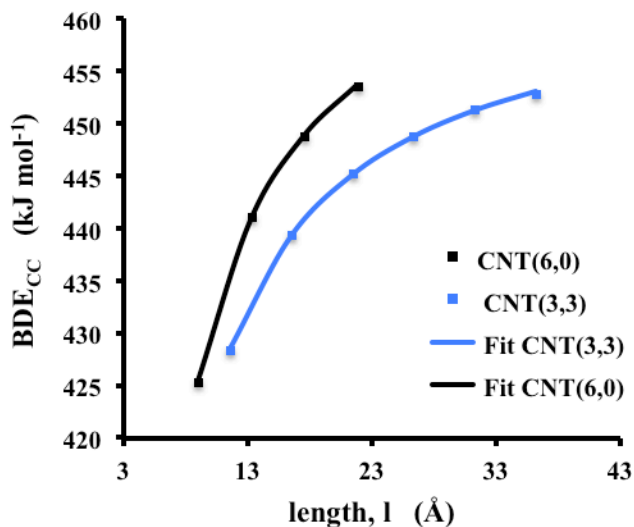


Fig. 3. Bond dissociation enthalpy per C-C bond as a function of the CNT length. The length of the CNT(6,0) and CNT(3,3) ranged between 9.052 Å - 21.841 Å and 11.585 Å - 36.197 Å, respectively.

3.2 Size-dependent properties of SWCNT as a function of radius

The radius of a CNT is a significant geometric parameter for CNTs as it determines the curvature of the hexagonal carbon sheet that is rolled into the tubular structure. The larger the radius or diameters of a CNT the flatter its surface. For CNTs with radius greater than 10 nm electronic properties similar to graphene would be expected. In Fig. 4 the BDE_{CC} is plotted as a function of $1/radius^2$ of various arm-chair (i.e. CNT(n,n) where n = 3, 4, 5, 6, 7) and zig-zag (i.e. CNT(m,0) where m = 6, 7, 8, 9, 10) chirality CNTs, respectively, with various lengths (i.e. 9.0 - 36.2 Å) and radii (i.e. 4.2 - 9.6 Å). The excellent fitting of the $1/radius^2$ to the BDE can be explained by similar fitting of the strain energies that has been previously been found in first principle calculations of the elastic properties of CNTs.[32] Similar $1/radius^2$ dependance of the strain energy was found in carbo nano-toroids by Goddard and co-worker.[33] The arm-chair CNTs have higher BDE_{CC} than the zig-zag CNT for lengths below 4 nm and radii under 1 nm. This suggests that arm-chair CNT have a larger Young's modulus, tensile strength, and bending modulus than zig-zag SWCNTs when their length is below 4 nm and their radius less than 1 nm.

The lines in Fig. 4 are perfectly linear based on the $R^2 = 0.999$ and $R^2 = 0.994$ for arm-chair and zig-zag, respectively. For an infinitely long SWCNT the arm-chair CNT(n,n) and zig-zag CNT(m,0) we find that the BDE_{CC} becomes 480 kJ mol⁻¹ and 479 kJ mol⁻¹, which means that

on the basis of radius both arm-chair and zig-zag have the same BDE_{CC} (taking into consideration the numerical limitations of the calculations) when their radius is large. Furthermore, at small radii (radius = 1 nm) the BDE_{CC} of arm-chair CNTs is similar to the BDE_{CC} of zig-zag CNTs to within 2 kJ mol^{-1} . Therefore, the BDE_{CC} of both arm-chair and zig-zag CNTs is the same as a function of their radius or curvature..

Also in Fig. 4 we observe that the BDE_{CC} becomes greater as the radius of the CNTs increases, which means that the carbon nanotube becomes thermodynamically more stable. We explain this increase of the thermodynamic stability via PDOS plots in section 2 which show that the overlap of the p orbitals of the π -system is greater when the curvature is smaller (i.e larger radii).

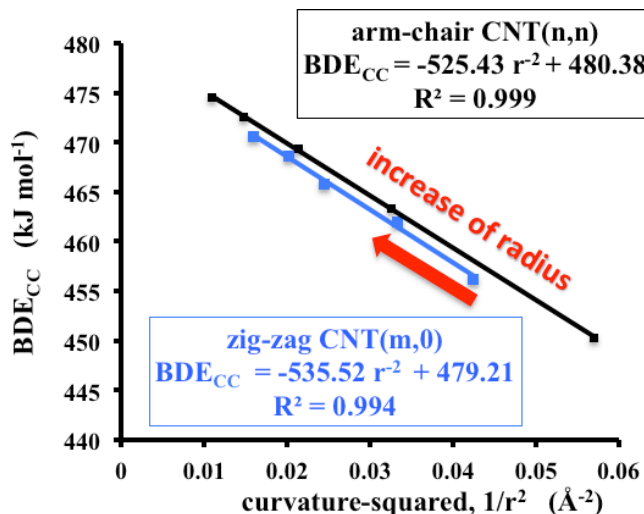


Fig. 4. Bond dissociation enthalpy per C-C bond as a function of $1/r^2$ for arm-chair CNT(n,n) and zig-zag CNT($m,0$), where $n = 3, 4, 5, 6, 7$ and $m = 6, 7, 8, 9, 10$. The radius of the CNT(n,n) and CNT($m,0$) ranged between $4.191 \text{ \AA} - 9.605 \text{ \AA}$ and $4.861 \text{ \AA} - 7.956 \text{ \AA}$, respectively.

In Fig. 5 we have plotted the BDE_{CC} as a function of the SWCNT radius in order to show the converging trend of this thermodynamic property. Similar to the conclusions reached from Fig. 4, we observe that a radius of 10 \AA is not sufficient to observe convergence of the BDE_{CC} . By fitting these curves to a function of the form $BDE_{CC} = A - B/r^2$ we have seen that at infinite radius, the BDE_{CC} of the arm-chair and zig-zag SWCNT become $480.2 \text{ kJ mol}^{-1}$ and $479.2 \text{ kJ mol}^{-1}$, respectively, which are identical within the accuracy of DFT calculations ($\sim 1\text{-}2 \text{ kJ mol}^{-1}$)

but we also find that the BDE_{CC} becomes constant, to within 0.1 kJ mol^{-1} , for SWCNT, with radii greater than 7.3 nm. Therefore, bundles of SWCNT that have radii greater than 7 nm will have isotropic elastic properties which is an important design feature in the use of CNTs in various applications.

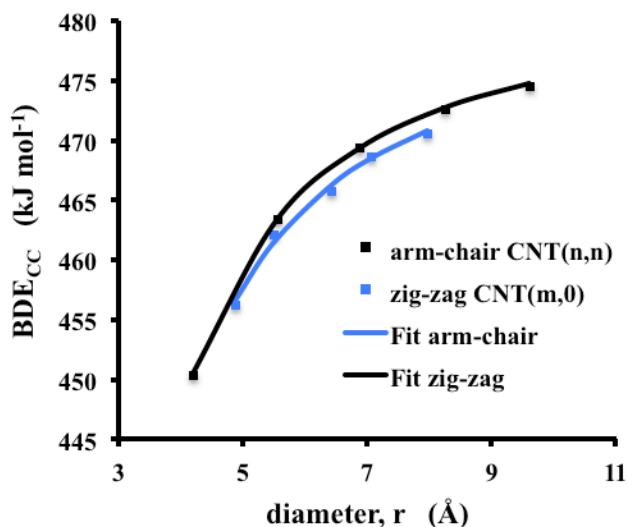


Fig. 5. Bond dissociation enthalpy per C-C bond as a function of the radius for arm-chair CNT(n,n) and zig-zag CNT(m,0), where $n = 3, 4, 5, 6, 7$ and $m = 6, 7, 8, 9, 10$. The radius of the CNT(n,n) and CNT(m,0) ranged between $4.191 \text{ \AA} - 9.605 \text{ \AA}$ and $4.861 \text{ \AA} - 7.956 \text{ \AA}$, respectively.

The trends in the BDE_{CC} of CNTs as a function of their diameter can be explained considering that there are certain energy requirements to bend a planar graphene sheet and roll it into a nanotube. We have previously studied what the bending modulus is in graphene nanoribbons and found that energy is required to bend a graphene nanoribbon (GNR), which suggests that carbon materials with curved surfaces have higher energies than flat GNR.[30] The bending energy is considerably smaller than the shearing and compression energy as we have previously shown for GNR of varying sizes.[30] When the GNR is bent there is less overlap between the $C p_z$ orbitals primarily, which causes the BDE_{CC} to decrease as the curvature of the CNT increases. Furthermore, the largest BDE_{CC} should be observed in flat graphene sheets.[28,34] This means that in MWCNT the outermost SWCNT will always have the largest elastic properties, such as Young's modulus, tensile strength, and bending modulus.

3.3 Partial-Density-of-States Plots of various SWCNTs

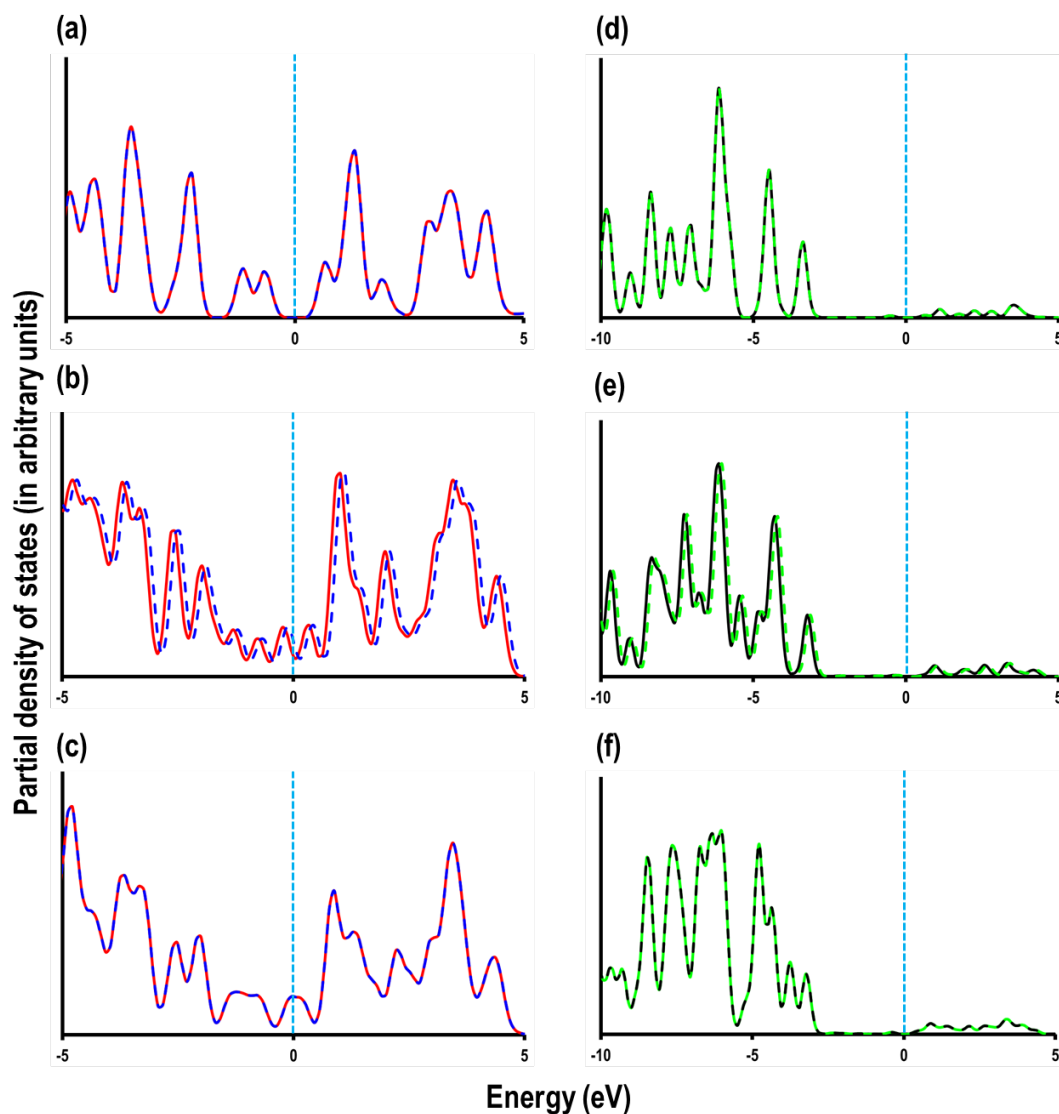


Fig. 6. Partial-density-of-states (PDOS) for the two carbon atoms in the centre of the CNT(3,3) with lengths of (a) 10 Å, (b) 20 Å and (c) 30 Å. The solid red (is the first carbon atom) and dotted blue (is the second carbon atom) lines, represents the summation of the contribution from C (p_x , p_y and p_z) orbitals of the two carbon atoms from the center of the CNT under consideration. The PDOS of the p_z orbitals (solid black line for the first carbon atom and the

green dotted line for the second carbon atom) of the CNT(3,3) with length of (d) 10Å, (e) 20 Å and (f) 30 Å. The vertical dotted line at zero represents the Fermi energy (E_F).

Density-of-states (DOS) have been previously used to explain the closing of the bandgap in SWCNTs as their dimensions increase.[12] However, in order to understand the size-dependent BDE trends of SWCNTs we have considered the Partial density of states (PDOS), which shows how the overlap between p electrons changes as a function of the SWCNT dimension. First, in order to understand the increase in the BDE_{CC} with increasing length of the SWCNT we consider the PDOS for the CNT(3,3) with length of 10Å, 20Å and 30Å (see Fig. 6 (a-c)). For these plots, we choose two carbon atoms from the center of the SWCNTs. The summation of p_x , p_y , and p_z contributions around the Fermi energy (E_F) show that with increasing length, the gap between the highest occupied and the lowest unoccupied states near the E_F decrease (Fig. 6 (a-c)). We also observe that for the SWCNT(3,3) with 30Å length there are $C p$ states at the E_F , meaning that this SWCNT is conducting while SWCNT(3,3) with a length of 10Å, and 20Å are semiconducting. To clarify the chemical bonding in these two carbon atoms we analyse the overlapping between their $C p_z$ orbitals. As shown in Fig. 6 (d-f), the $C p_z$ orbitals, as expected, are completely overlapped with each other but with the increasing length of the SWCNT(3,3) these overlapping $C p_z$ orbital signatures become broader. In particular, the area under the PDOS plots from -5.5 eV to -2.7 eV was found to be 0.55 eV, 0.57 eV and 0.63 eV for the 10Å, 20Å and 30Å SWCNT(3,3), respectively. An increased overlap between p_z orbitals means that the C–C bonds become stronger as the length of the CNT increases. This also explains the increase in BDE_{CC} with the increase in the SWCNT lengths. This observation has significant applications to it as it suggests that CNT-composite materials that have the same $\text{mass}(\text{CNT})/\text{mass}(\text{composite})$ the one with the lengthier CNTs will have in principle larger Young's modulus, tensile strength and bending modulus.

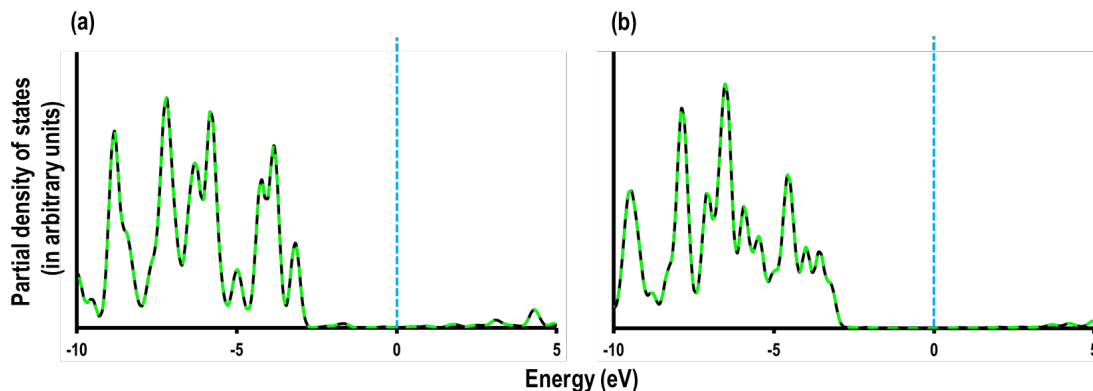


Fig. 7. Partial density of states for the p_z orbitals of the two carbon atoms from the centre of the (a) SWCNT(6,0) and (b) SWCNT(9,0) with a length of 25 Å. The solid black line is the p_z orbital signature for the first carbon atom and the green dashed line for the second carbon atom. The vertical dashed line at zero represents the Fermi energy (E_F).

In Fig. 7 we explore the effect of the radius of a SWCNT on the BDE_{CC} by considering the PDOS of the SWCNT(6,0) and SWCNT(9,0) with a similar length of 25Å. The analysis on p_z orbital in these systems show that as the radius increases the overlap between the p_z -orbitals increases significantly, which is evident from the broad overlapping signatures of SWCNT(9,0) as compared to SWCNT(6,0). The reason for the increased overlap between the p_z -orbitals in SWCNT(9,0) is related to the decrease in curvature, which is comparatively more pronounced in SWCNT(6,0). We conclude that the increase in BDE_{CC} with increasing radius of SWCNT is related to the greater p_z - p_z overlap and hence stronger chemical bonding between the carbon atoms. This is in agreement with a previous study where we have evaluated the BDE for various carbon materials (e.g. fullerene, carbon nanocones, CNTs, graphene nanoribbon) and found the BDE increases as the carbon material becomes flatter.[28,34]

4. Conclusions

We present a detailed DFT study of the thermodynamic properties of Single-Walled Carbon Nanotubes of various lengths, diameters and chirality. Our study shows that the BDE_{CC} is inversely proportional to the curvature-squared ($1/r^2$) and inversely proportional of the SWCNT length ($1/l$). We derive quantitative relationships from which the BDE can be calculated as a function of size and radius of the SWCNT. This suggests that SWCNT have stronger elastic properties as their length increases due to the emergence of metallic properties and as their radius

increases due to larger overlap of the p orbitals. This clearly shows that there is a size-effect of the thermodynamic properties of SWCNTs when their length and diameter is less than 4 and 1 nm, respectively, which should be considered in the design of materials that use SWCNT of such dimensions. Lastly, we calculate that the BDE in SWCNTs is intermediate between the BDE of a C-C and C=C bond, confirming partial double bond of the carbon framework.

Conflicts of interest

There are no conflicts to declare.

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Supplementary Material

The coordinates of optimised SWCNT are given as supporting information (Fig. S1).

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